

Научном већу Института за физику

Београд, 07.03.2018.

Предмет:

Покретање поступка за избор у звање истраживач сарадник

Молим Научно веће Института за физику да покрене поступак за мој избор у звање истраживач сарадник.

У прилогу достављам:

1. мишљење руководиоца пројекта са предлогом комисије за избор у звање
2. стручну биографију
3. преглед научне активности
4. списак објављених научних радова и других публикација
5. копије објављених радова
6. уверење о последњем уписаном и овереном семестру на докторским студијама
7. фотокопију дипломе
8. фотокопију индекса са докторских студија
9. потврду Колегијума докторских студија Физичког факултета о прихватању предлога теме докторске дисертације;

С поштовањем,

Ана Милосављевић

Научном већу Института за физику

Београд, 02.03.2018.

Предмет:

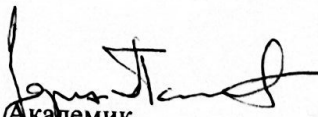
Мишљење руководиоца пројекта о избору Ане Милосављевић у звање истраживач сарадник

Ана Милосављевић запослена је у Центру за физику чврстог стања и нове материјале, Института за физику и ангажована на пројекту Министарства просвете, науке и технолошког развоја Републике Србије III45018 "Наноструктурни мултифункционални материјали и нанокompозити". На пројекту ради на темама везаним за динамику решетке суперпроводника на бази гвожђа под руководством др Ненада Лазаревића, вишег научног сарадника Института за физику. Коаутор је два научна рада која су објављена у врхунским међународним часописима (видети прилог). С обзиром да испуњава све услове у складу са Законом о научноистраживачкој делатности и Правилником о поступку, начину вредновања и квалитативном исказивању научноистраживачких резултата истраживача МПНТР, сагласан сам са покретањем избора Ане Милосављевић у звање истраживач сарадник.

За састав комисије за избор Ане Милосављевић у звање истраживач сарадник предлажем:

1. др Ненада Лазаревића, вишег научног сарадника Института за физику,
2. др Мају Шћепановић, научног саветника Института за физику,
3. др Милоша Радоњића, научног сарадника Института за физику,
4. др Божидара Николића, доцента Физичког факултета Универзитета у Београду

Руководилац пројекта III45018



Академик
Зоран В. Поповић

Биографски подаци

Ана Милосављевић рођена је 30.06.1986. године у Смедереву. Након завршене основне школе и гимназије, 2005. године уписује основне студије на Математичком факултету Универзитета у Београду, на ком је дипломирала 2010. године. Исте године уписује мастер студије на Физичком факултету Универзитета у Београду, смер Теоријска и експериментална физика. 2013. године брани мастер рад под називом *"Вибрационе особине ZnO нанотуба"* и тиме завршава мастер студије са просечном оценом 9,33. Исте године уписује докторске студије на Физичком факултету Универзитета у Београду, ужа научна област – Физика кондензоване материје и статистичка физика. Од 10.03.2015. године запослена је у Центру за физику чврстог стања и нове материјале Института за физику, као истраживач приправник на пројекту Министарства просвете, науке и технолошког развоја Републике Србије III45018 "Наноструктурни мултифункционални материјали и нанокомпозити", чији је руководилац академик Зоран В. Поповић.

Преглед научне активности Ане Милосављевић

Ана Милосављевић се у свом научном раду бави динамиком решетке суперпроводника на бази гвожђа, користећи методу Раманове спектроскопије. Међу овим материјалима посебна пажња усмерена је на класу гвожђе халкогенида. Гвожђе халкогениди су изоструктурни и изоелектронски материјали, веома различити по својим физичким особинама. Од члана до члана, карактеристике као што су структурни фазни прелаз, електронска нематичност, магнетно уређење и суперпроводно стање веома се разликују. Сматра се да у случају FeSe, првог синтетисаног суперпроводника из ове класе, електронска нематичност има утицаја на механизам формирања Куперових парова а експерименти показују да ово стање постоји и у $\text{FeSe}_{1-x}\text{S}_x$, $0 \leq x \leq 0.17$. Испитивањем утицаја нематичне фазе посредством фонона, могао би да се стекне увид у начин на који она има утицаја на формирање Куперових парова.

У оквиру активности везане за ову проблематику, Ана Милосављевић се бавила испитивањем динамике решетке FeS, најновијег члана фамилије гвожђе халкогенида. Резултати експеримената Рамановог расејања на овом материјалу, омогућили су идентификацију два фононска мода (A_{1g} и B_{1g}), али и трећег фононског мода који се према теоријским прорачунима налази у процепу фононске густине стања. За овај фононски мод утврђено је да потиче од расејања другог реда, што је на основу селекционих правила за двофононске процесе добијених методом модификованих групних пројектора потпуно потврђено. Идентификован је четврти мод у непосредној близини A_{1g} мода, који потиче или од дефектом индукованог расејања или који је, као и у случају предходног мода, резултат двофононског процеса. Установљено је да је температурска зависност свих модова између 300 и 20 К, потпуно одређена контракцијама решетке. Између 20 и 4 К енергије свих модова трпе скок, док полуширина A_{1g} мода опада а B_{1g} мода расте. Овакво понашање фонона индикација је кратकोдометног магнетног уређења. Појава двофононског процеса индикација је јаке фонон-фонон интеракције у овом материјалу, која највероватније потиче од појачане електрон-фонон интеракције у односу на остале чланове класе гвожђе халкогенида.

Списак публикација Ане Милосављевић

Публикације у врхунским међународним часописима (категирија М21)

1. A. Baum, **A. Milosavljević**, N. Lazarević, M. M. Radonjić, B. Nikolić, M. Mitschek, Z. Inanloo Maranloo, M. Šćepanović, M. Grujić-Brojčin, N. Stojilović, M. Opel, A. Wang, C. Petrovic, Z. V. Popović and R. Hackl, "Phonon anomalies in FeS", Phys. Rev. B 97, 054306 (2018)
2. M. Орачић, N. Lazarević, D. Tanasković, M.M. Radonjić, **A. Milosavljević**, YongChang Ma, C. Petrovic and Z. V. Popović, "Small influence of magnetic ordering on lattice dynamics in TaFe_{1,25}Te₃", Phys. Rev. B 96, 174303 (2017)

Саопштења са међународног скупа штампана у изводу (категирија М34)

1. M. Орачић, N. Lazarević, D. Tanasković, M.M. Radonjić, **A. Milosavljević**, YongChang Ma, C. Petrovic and Z. V. Popović, "Small influence of magnetic ordering on lattice dynamics in TaFe_{1,25}Te₃", Sixteenth young researchers' conference materials science and engineering, Belgrade, Serbia, 6. - 8. December 2017, Book of Abstracts, p. 42 (2017)
2. **A. Milosavljević**, A. Baum, N. Lazarević, M. M. Radonjić, B. Nikolić, M. Mitschek, Z. Inanloo Maranloo, M. Šćepanović, M. Grujić-Brojčin, N. Stojilović, M. Opel, A. Wang, C. Petrovic, Z. V. Popović and R. Hackl, "Phonon anomalies in FeS", Sixteenth young researchers' conference materials science and engineering, Belgrade, Serbia, 6. - 8. December 2017, Book of Abstracts, p. 44 (2017)

Small influence of magnetic ordering on lattice dynamics in TaFe_{1.25}Te₃M. Opačić,¹ N. Lazarević,¹ D. Tanasković,² M. M. Radonjić,² A. Milosavljević,¹ Yongchang Ma,^{3,4}
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Raman scattering spectra of zigzag spin chain TaFe_{1.25}Te₃ single crystal are presented in a temperature range from 80 to 300 K. Nine Raman active modes of A_g and B_g symmetry are clearly observed and assigned by probing different scattering channels, which is confirmed by lattice dynamics calculations. Temperature dependence of the Raman modes linewidth is mainly governed by the lattice anharmonicity. The only deviation from the conventional behavior is observed for A_g symmetry modes in a vicinity of the magnetic phase transition at $T_N \approx 200$ K. This implies that the electron-phonon interaction weakly changes with temperature and magnetic ordering, whereas small changes in the spectra near the critical temperature can be ascribed to spin fluctuations.

DOI: [10.1103/PhysRevB.96.174303](https://doi.org/10.1103/PhysRevB.96.174303)**I. INTRODUCTION**

The discovery of superconductivity in La(O_{1-x}F_x)FeAs in 2008 [1] initiated an intensive search for new iron-based superconducting materials, in order to obtain better understanding of their physical properties and the mechanism of high- T_c superconductivity [2–4]. Novel iron-based materials, however, are not only superconducting, but can also exhibit various types of magnetic ordering. In some cases the magnetic phase transition is continuous [5–8], whereas in others it is accompanied by structural changes [9–15], or even by a nanoscale coexistence of antiferromagnetic (AFM) and superconducting domains [16–18].

TaFe_{1+y}Te₃ was synthesized and characterized about 25 years ago [19,20]. It is a layered system consisting of FeTe chains, along the b axis, separated by a Ta/Te network in between; see Fig. 1. These layers are parallel to the natural cleavage plane. There are also additional Fe ions, Fe2, randomly occupying interstitial sites [21–23]. TaFe_{1+y}Te₃ features anisotropic charge transport with metallic resistivity within the plane and insulating in the direction normal to the FeTe layers [23]. The first study of magnetic structure implies that TaFe_{1+y}Te₃ is composed of double zigzag spin chains with antiferromagnetic ordering of Fe1 spins [22]. The newest neutron diffraction measurements suggest that spin ordering within zigzag chains is ferromagnetic, whereas these zigzag chains couple antiferromagnetically [23], as shown in Fig. 1(b). However, the exact interaction mechanism is not clearly resolved.

There is a similarity between TaFe_{1+y}Te₃ and the extensively studied Fe_{1+y}Te compound since they are correlated bad metals which order antiferromagnetically below $T_N \approx 200$ K and 70 K, respectively [10,23], both having rather large magnetic moments on Fe ions, $\approx 2 \mu_B/\text{Fe}$. TaFe_{1+y}Te₃, however, forms ferromagnetic (FM) zigzag spin chains which couple antiferromagnetically between the layers, whereas the Fe spins in Fe_{1+y}Te form a bicollinear AFM structure. The magnetic phase transition in Fe_{1+y}Te is accompanied by the

structural change from a tetragonal to a monoclinic, as opposed to TaFe_{1+y}Te₃ where a continuous transition to the AFM phase is observed in thermodynamic and transport measurements [22]. Just like in Fe_{1+y}Te, interest in spin chain and ladder materials [24] stems not only from their block-AFM states similar to parent compounds of iron-based superconductors [25], but also from superconductivity. It is worth noting that spin 1/2 copper oxide ladder structures host a spin gap and superconductivity upon doping [26–28]. In contrast to superconductivity in copper oxide ladder materials that was rather rare and with critical temperatures rather small when compared to highest achieved in copper square lattices [29,30], iron-ladder materials feature T_c 's similar to the highest found in Fe-based superconductors [31].

Raman spectra provide additional information on magnetic ordering and electron-phonon coupling. There exist several Raman studies of the phonon spectra of iron based materials near the superconducting or magnetic phase transition [32,33]. While no anomalies were observed in 1111 compounds [34,35], the Raman spectra show anomalous behavior near the spin density wave (SDW) transition in some of the 122 and 11 compounds [15,36–38], which was ascribed to the phonon renormalization due to the opening of the SDW or superconducting gap, or to the structural transition. Large anomalies were observed also in ferromagnetic K_xCo_{2-y}Se₂ [5], which was ascribed to the effect of electron-phonon coupling and spin fluctuations. Fe_{1+y}Te phonon spectra feature unusually large anomalies near the magnetic phase transition, as seen in sudden changes in the phonon frequencies and linewidths, due to the phonon modulation of magnetic interactions and structural phase transition [11–13]. Therefore, it is of interest to examine lattice dynamics in the normal state of iron-spin chain and ladder materials and compare it to materials like Fe_{1+y}Te. To the best of our knowledge, there are no published data on lattice dynamics of TaFe_{1+y}Te₃.

In this paper we present polarized Raman scattering spectra of TaFe_{1.25}Te₃ single crystal measured in a temperature range from 80 to 300 K. Nine out of 15 Raman active modes are

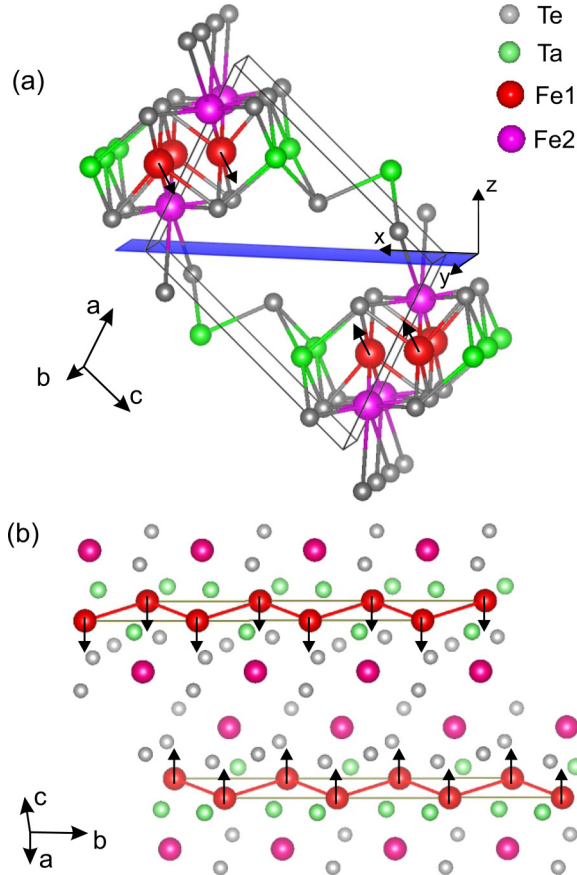


FIG. 1. (a) Structure of the $\text{TaFe}_{1.25}\text{Te}_3$ single crystal together with the natural cleavage plane $[\bar{1}01]$. $\mathbf{x} = 1/\sqrt{2}(\bar{1}0\bar{1})$ and $\mathbf{y} = (010)$ correspond to our laboratory system. (b) A view of the $\text{TaFe}_{1.25}\text{Te}_3$ structure along the b axis. Two neighboring chains of Fe1 spins point in a parallel direction, forming a ferromagnetic zigzag chain, whereas spins of neighboring zigzag chains order antiferromagnetically. One should note that Fe2 is present with occupancy of 0.25.

observed and assigned using the selection rules for different polarization configurations and lattice dynamics calculations. In a sharp contrast to the related FeTe compound, $\text{TaFe}_{1.25}\text{Te}_3$ Raman spectra do not show significant changes near $T_N \approx 200$ K, which clearly indicates that the phase transition is continuous. Temperature dependence of the frequency and linewidth is conventional, driven by the anharmonicity effects, except very near T_N where some of phonon lines slightly broaden which should be the consequence of spin fluctuations near the critical temperature. These results indicate very small changes in the electron-phonon coupling and in the Fermi surface in the measured temperature range.

II. EXPERIMENT AND NUMERICAL METHOD

Single crystals were grown using the self-flux method, as described elsewhere [19]. Raman scattering measurements were performed on freshly cleaved $(\bar{1}01)$ -oriented samples, using Jobin Yvon T64000 Raman system, equipped with a nitrogen-cooled CCD detector, in the backscattering micro-Raman configuration. The 532 nm line of a solid state laser was used as an excitation source. A microscope objective with

50 \times magnification was used for focusing the laser beam. All measurements were performed at low laser power, to reduce local heating of the sample. For low temperature measurements KONTI CryoVac continuous flow cryostat with 0.5 mm thick window was used. All spectra were corrected for the Bose factor. For extracting the data from the Raman spectra, phonon modes were fitted with a Lorentzian profile.

The electronic structure is calculated for stoichiometric TaFeTe_3 in the paramagnetic phase within the density functional theory (DFT), and the phonon frequencies at the Γ -point are obtained within the density functional perturbation theory (DFPT) [39], using the QUANTUM ESPRESSO package [40]. We have used projector augmented wave (PAW) pseudopotentials with Perdew-Burke-Ernzerhof (PBE) exchange-correlation functional with nonlinear core correction and Gaussian smearing of 0.01 Ry. The electron wave function and the density energy cutoffs were 64 Ry and 782 Ry, respectively. The Brillouin zone is sampled with $8 \times 8 \times 8$ Monkhorst-Pack k -space mesh. The phonon frequencies were calculated with the unit cell size taken from the experiments and the relaxed positions of atoms within the unit cell. The forces acting on individual atoms in the relaxed configuration were smaller than 10^{-4} Ry/a.u.

III. RESULTS AND DISCUSSION

$\text{TaFe}_{1+y}\text{Te}_3$ crystallizes in the monoclinic crystal structure, which is shown in Fig. 1. The space group is $P2_1/m$ (unique axis b), with two formula units per unit cell [19,20]. The experimental values of the unit cell parameters are $a = 7.436$ Å, $b = 3.638$ Å, $c = 10.008$ Å, and $\beta = 109.17^\circ$. All atoms (including the excess Fe), are at $2e$ Wyckoff positions, with fractional coordinates given in Table I.

The factor group analysis (FGA) for $P2_1/m$ space group yields the following normal mode distribution at the Γ point:

$$\Gamma_{\text{Raman}} = 10A_g + 5B_g,$$

$$\Gamma_{\text{IR}} = 4A_u + 8B_u,$$

$$\Gamma_{\text{acoustic}} = A_u + 2B_u.$$

The Raman spectra were measured from the $(\bar{1}01)$ plane of the sample, which is the natural cleavage plane [23,42]. From the Raman tensors given in Table II, the A_g phonon modes are expected to be observable in the (xx) and (yy) scattering configurations. The B_g modes can be observed only in (xy) polarization configuration.

TABLE I. Experimental fractional coordinates of $\text{TaFe}_{1.25}\text{Te}_3$ taken from Ref. [19]. In the parentheses are the calculated values for TaFeTe_3 .

Atom type	x	y	z
Ta	0.8340 (0.8331)	0.25	0.3007 (0.2987)
Fe1	0.6147 (0.6223)	-0.25	0.0890 (0.0988)
Fe2	0.7686	0.25	-0.0047
Te1	0.4392 (0.4326)	0.25	0.1860 (0.1637)
Te2	0.9835 (0.9842)	-0.25	0.1589 (0.1584)
Te3	0.2179 (0.2192)	0.25	0.4970 (0.5028)

TABLE II. Upper panel: atomic species (all of them are at $2e$ Wyckoff positions) and the contribution of each atom to the Γ -point phonons, the corresponding Raman tensors for the TaFeTe₃ single crystal ($P2_1/m$ space group) [41]. Lower panel: the calculated (for the stoichiometric TaFeTe₃) and experimental phonon energies at 100 K (for the TaFe_{1.25}Te₃ single crystal).

Atoms			Irreducible representations		
Ta, Fe1, Te1, Te2, Te3			$2A_g + A_u + B_g + 2B_u$		
$\hat{R}_{A_g} = \begin{pmatrix} a & 0 & d \\ 0 & b & 0 \\ d & 0 & c \end{pmatrix}$			$\hat{R}_{B_g} = \begin{pmatrix} 0 & e & 0 \\ e & 0 & f \\ 0 & f & 0 \end{pmatrix}$		
Raman active			Infrared active		
Symmetry	Calc. (cm ⁻¹)	Expt. (cm ⁻¹)	Symmetry	Calc. (cm ⁻¹)	Expt. (cm ⁻¹)
A_g^1	36.2		A_u^1	42.8	
B_g^1	43.8		B_u^1	54.9	
B_g^2	57.9	61.6	B_u^2	94.4	
A_g^2	63.8	62.3	A_u^2	101.4	
A_g^3	75.3	68.5	B_u^3	111.3	
A_g^4	104.4	90	A_u^3	131.1	
B_g^3	105.1		B_u^4	143.2	
A_g^5	124.6		B_u^5	160.4	
B_g^4	127.2	130.4	B_u^6	188.6	
A_g^6	149.8	155	B_u^7	227.9	
A_g^7	164.9	165	A_u^4	231.1	
A_g^8	191		B_u^8	289.4	
B_g^5	217.1	222.3			
A_g^9	241.9	223.9			
A_g^{10}	276.22				

Raman scattering spectra of TaFe_{1.25}Te₃ single crystals, measured at 100 K in three different polarization configurations, are presented in Fig. 2. By using the selection rules, we assign the Raman peaks appearing in the (xx) and (yy) polarization configuration as the A_g ones. This conclusion is supported by the lattice dynamics calculations, given in Table II. By comparing the calculated values of A_g mode energies with those of the peaks appearing in the (xx) and (yy) spectra, we can unambiguously assign four Raman modes (A_g^4 , A_g^6 , A_g^7 , and A_g^9). The broad structure around 65 cm⁻¹ probably originates from the A_g^2 and A_g^3 modes, although the contribution of the A_g^1 mode (with calculated energy of 42.7 cm⁻¹) cannot be excluded. The peaks at 57.9 cm⁻¹ and 130 cm⁻¹ that are clearly visible in (xy) but absent in (yy) configuration are assigned as B_g^2 and B_g^4 modes, respectively. The low intensity peak at ≈ 220 cm⁻¹, that becomes clearly observable at low temperatures, is tentatively assigned as B_g^5 mode, although the contribution from the leakage of A_g^9 mode cannot be excluded. The origin of the two very broad structures at about 70 cm⁻¹ and 160 cm⁻¹, which are pronounced in the (xy) configuration, is not completely clear. Aside from providing additional charge, Fe2 atoms may contribute to momentum transfer scattering, in line with the pronounced quasielastic continuum, present in all the scattering configurations. Consequently, contribution from single-phonon scattering away from Γ point becomes observable, which is theoretically predicted [43,44] and experimentally observed

[45,46]. Although we cannot exclude the possibility of two- and, in particular, double-phonon contributions, we believe it is less likely due to the nature of the processes and since they usually have more pronounced contribution to A channel (for arbitrary irreducible symmetry μ of C_{2h} holds $\mu \otimes \mu \ni A$).

The normal modes of the selected A_g and B_g vibrations, as obtained by the lattice dynamics calculations, are shown in Fig. 3. The low energy B_g^2 mode represents vibrations of Te and Ta atoms which tend to elongate the (Ta,Fe)Te tetrahedra in the xy plane. B_g^4 phonon originates mainly from Ta and Te atom vibrations in directions opposite to each other, whereas A_g^4 mode represents dominantly vibrations of another Te atom in the xy plane. A_g^7 and A_g^9 modes originate from the vibrations of Fe and Te atoms which tend to rotate the tetrahedra around the x axis.

The DFT calculations are in very good agreement with the measured Raman spectra, specially having in mind the strength of electronic correlations in iron based compounds and the presence of additional Fe2 atoms in the measured sample. We restricted to the nonmagnetic DFT calculations. This is because small changes in the phonon energies due to the magnetic ordering cannot be reliably captured since the DFT does not treat strong correlation and spin-fluctuations effects. Our DFT calculations for the electronic band structure agree with the results from Ref. [42]. The calculated electronic dispersions are in rather good agreement with the ARPES measurements [42], which indicates that the main effect of the interstitial Fe2 ion is to provide additional charge and shift

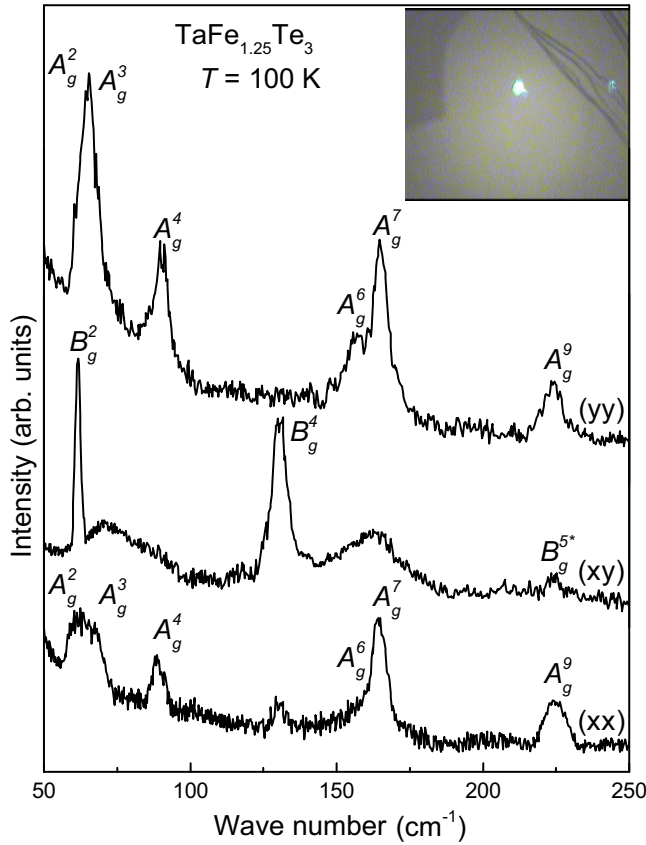


FIG. 2. Polarized Raman scattering spectra of $\text{TaFe}_{1.25}\text{Te}_3$ single crystal measured at 100 K in various polarizations. The notation in parentheses indicates the polarization directions of the incident and scattered light according to Fig. 1(a). Inset: surface of the probed $\text{TaFe}_{1.25}\text{Te}_3$ single crystal.

the Fermi level. This conclusion is supported with a small difference between the relaxed and experimental fractional coordinates; see Table I.

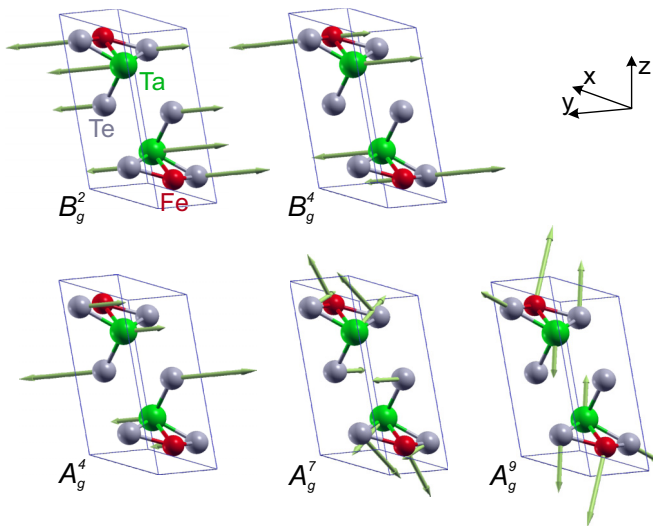


FIG. 3. Unit cell of TaFeTe_3 single crystal with the displacement patterns of several A_g and B_g Raman modes. Arrow lengths are proportional to the square root of the interatomic forces.

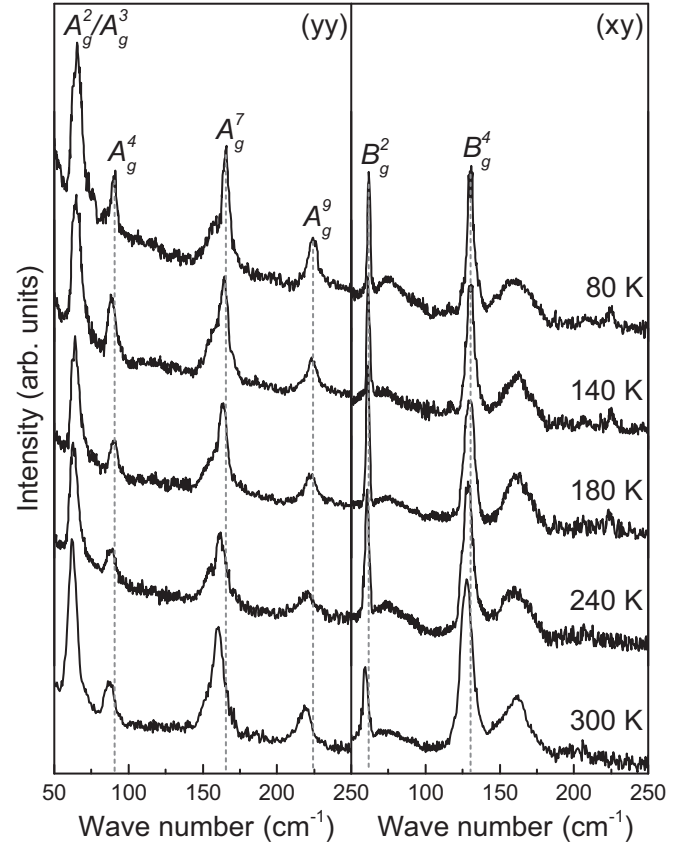


FIG. 4. Temperature dependent Raman scattering spectra of $\text{TaFe}_{1.25}\text{Te}_3$ single crystal in the (yy) (left panel) and (xy) (right panel) polarization.

In order to analyze the changes of the Raman spectra near the AFM transition at $T_N \approx 200$ K, we have performed measurements in a temperature range from 80 K up to 300 K. Raman spectra of $\text{TaFe}_{1.25}\text{Te}_3$ single crystal, measured at different temperatures in the (yy) and (xy) scattering configurations, are given in Fig. 4. In the following, we perform the temperature analysis of the energy and the linewidth for five most clearly observed modes.

The temperature dependence of the Raman mode energy is usually described with [47,48]

$$\omega_i(T) = \omega_{0,i} + \Delta_i^V(\gamma_i, \alpha_i(T)) + \Delta_i^A(C_i), \quad (1)$$

where $\omega_{0,i}$ is a temperature independent contribution to the Raman mode energy. The second term represents a change of the phonon energy induced by the lattice thermal expansion and depends on the Grüneisen parameter γ_i and the thermal expansion coefficient $\alpha_i(T)$. The term Δ_i^A describes the anharmonicity induced change of the Raman mode energy which is a function of the anharmonic constant C_i . Both Δ_i^V and Δ_i^A have qualitatively the same temperature dependence. Since there are no reported experimental data on the temperature dependence of the lattice parameters for $\text{TaFe}_{1+y}\text{Te}_3$, we didn't attempt to fit the data, and the black dotted lines in Figs. 5 and 6 are guides to the eye. The $\omega_i(T)$ curves follow the “standard” [5,15,37,49,50] continuous decrease in energy with temperature, with very small anomalies near T_N except for the A_g^4 mode.

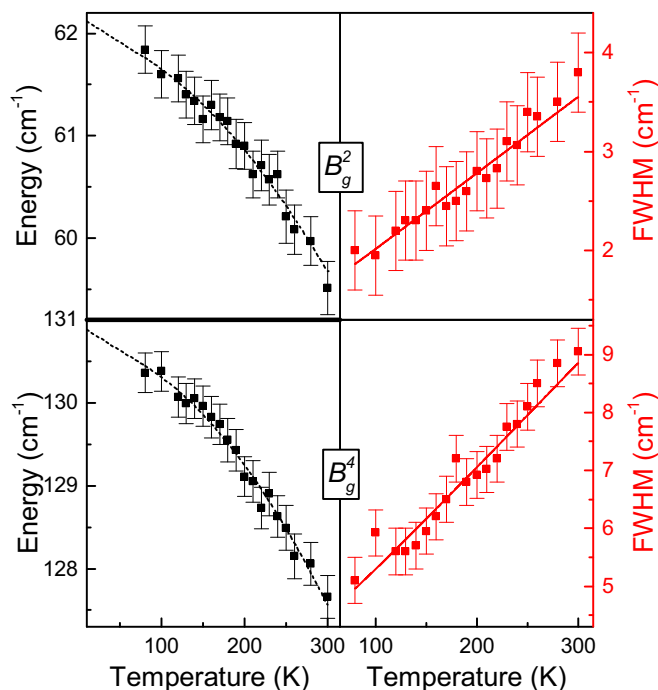


FIG. 5. Temperature dependence of the energy and linewidth for the B_g^2 and B_g^4 Raman modes of the $\text{TaFe}_{1.25}\text{Te}_3$ single crystal. The red lines are fitted according to Eq. (2), whereas black lines are guides to the eye.

The temperature dependences of the linewidth of selected B_g and A_g modes are given in the right panels of Figs. 5 and 6, respectively. While the B_g^2 and B_g^4 phonon modes do not show significant deviation from the usual behavior due to the anharmonicity effects, with gradual broadening with increasing temperature, the A_g^4 , A_g^7 , and A_g^9 modes exhibit moderate additional broadening above 200 K. The red lines present a fit to the standard formula for the temperature dependent linewidth due to the anharmonicity [11,47,51]:

$$\Gamma_i(T) = \Gamma_{0,i} \left(1 + \frac{2}{e^{\hbar\omega_{0,i}/2k_B T} - 1} \right) + A_i, \quad (2)$$

where $\Gamma_{0,i}$ is the anharmonic constant and A_i is the constant term due to the disorder and electron-phonon interaction [52]. The deviation from these anharmonicity curves is most pronounced around T_N (see the insets of Fig. 6).

We can observe that all Raman modes have moderate linewidth and exhibit small anomalies near T_N . This shows that the phase transition is continuous, in agreement with the thermodynamic and transport measurements [22]. Small anomalies in the phonon spectra, which are restricted only to the vicinity of the phase transition, imply that the electron-phonon interaction of Raman active modes does not change with temperature. This is in agreement with the recent ARPES measurements which show negligible change of the Fermi surface across the AFM transition [42], indicating that the magnetic transition is not driven by the Fermi surface instability. The anomalies in the linewidth of some phonon modes near T_N are likely the signature of the increased scattering by spin fluctuations near the phase transition [51,53].

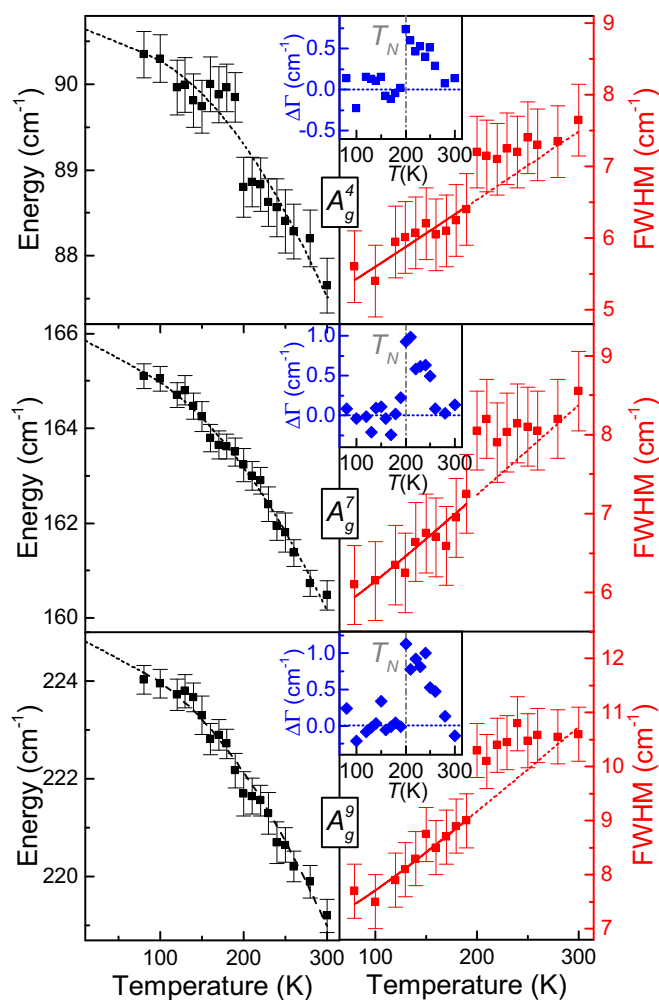


FIG. 6. Energy and linewidth of the A_g^4 , A_g^7 , and A_g^9 Raman modes of the $\text{TaFe}_{1.25}\text{Te}_3$ single crystal as a function of temperature. The red lines are plotted according to Eq. (2), and the black dotted lines are guides to the eye. The insets represent deviations of the Raman mode linewidth from the anharmonic form.

The density of states (DOS) at the Fermi level is not large. This can be concluded from the ARPES experiments [42] which have shown three bands crossing the Fermi level but with strong dispersion, while several relatively flat bands are found only well below the Fermi level. The DFT calculations also give moderate values for the DOS, $N(E_F) \approx 1 \text{ eV}^{-1}/\text{f.u.}$, after the Fermi level is shifted due to the additional charge provided by the Fe2 atoms. This value for the DOS also suggests that the electron-phonon coupling is not strong in $\text{TaFe}_{1.25}\text{Te}_3$, since it is proportional to $N(E_F)$.

$\text{TaFe}_{1.25}\text{Te}_3$ has a similar moment size as Fe_{1+y}Te , $\approx 2\mu_B/\text{Fe}$. However, the differences in the magnetic ordering and crystal structure cause different phonon properties of these two compounds. Namely, the phonon lines in the Raman spectra of Fe_{1+y}Te have very large linewidth and pronounced anomalies both in the frequency and in the linewidth near the first order phase transition [11,13]. Small anomalies in the Raman spectra of $\text{TaFe}_{1.25}\text{Te}_3$ as compared to Fe_{1+y}Te can be ascribed to the continuous, second order nature of the AFM transition and smaller electron-phonon coupling due to lower DOS at the

Fermi level. Also, the monoclinic angle β in the $\text{TaFe}_{1.25}\text{Te}_3$ unit cell significantly differs from 90° and therefore the form of the vibrational modes is different.

IV. CONCLUSION

In summary, we have performed the Raman scattering study of the zigzag spin chain $\text{TaFe}_{1.25}\text{Te}_3$ single crystal, together with the lattice dynamics calculations of TaFeTe_3 . By analyzing the Raman spectra in different polarization configurations and using numerical calculations we have assigned nine Raman active modes predicted by the FGA. Very good agreement between the experimental frequencies and those calculated for the stoichiometric compound shows that the excess iron atoms weakly influence the phonon energies but provide momentum conservation for the phonon scattering away from Γ point. The temperature dependence of the frequency and the linewidth of the B_g Raman modes looks conventional, governed by the anharmonicity effects. While in a broad temperature range the behavior of the A_g modes is also conventional, there are clear anomalies near

the AFM transition. The anomalies in the frequency and the linewidth are in the form of small kinks near T_N . This implies that the electron-phonon interaction and the DOS at the Fermi level are approximately constant in the measured temperature range. The increase in the linewidth near T_N is likely due to the coupling of spin fluctuations and vibration near the second order phase transition.

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Phonon anomalies in FeS

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We present results from light scattering experiments on tetragonal FeS with the focus placed on lattice dynamics. We identify the Raman active A_{1g} and B_{1g} phonon modes, a second order scattering process involving two acoustic phonons, and contributions from potentially defect-induced scattering. The temperature dependence between 300 and 20 K of all observed phonon energies is governed by the lattice contraction. Below 20 K the phonon energies increase by $0.5\text{--}1\text{ cm}^{-1}$, thus indicating putative short range magnetic order. Along with the experiments we performed lattice-dynamical simulations and a symmetry analysis for the phonons and potential overtones and find good agreement with the experiments. In particular, we argue that the two-phonon excitation observed in a gap between the optical branches becomes observable due to significant electron-phonon interaction.

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I. INTRODUCTION

In the iron based superconductors (IBS) magnetic order, structure, nematicity, and superconductivity are closely inter-related. Upon substituting atoms in the parent compounds the properties change in a way that the shape of the Fermi surface is generally believed to play a crucial role. Yet, the magnetic properties were found recently to be more complex and to depend also on the degree of correlation in the individual d orbitals contributing to the density of states close to the Fermi surface [1–3].

The influence of correlation effects seems to increase from the 122 systems such as BaFe_2As_2 to the 11 chalcogenides FeTe , FeSe , and FeS [4,5]. Surprisingly, the properties of the 11 class members differ substantially although they are isostructural and isoelectronic [3,6]: FeSe undergoes a structural transition at $T_s \sim 90\text{ K}$ and displays electronic nematicity [7]. While long-range magnetic order cannot be observed down to the lowest temperatures [7–10] the thermodynamic properties and the Raman spectra strongly support the presence of short-ranged magnetism [11,12]. Below $T_c \sim 9\text{ K}$ superconductivity is observed [13] in pristine FeSe . In mono-layer FeSe T_c can reach values close to 100 K [14,15].

The replacement of Se by Te leads to slightly off-stoichiometric Fe_{1+y}Te which exhibits a simultaneous magnetostructural transition near 67 K [16] but is not superconducting [17,18]. Finally, FeS having a superconducting transition at $T_c \sim 5\text{ K}$ [19] remains tetragonal down to the lowest temperatures [20]. It is still an open question whether tetragonal FeS hosts magnetic order. Obviously, the iron-chalcogenides are at the verge of various neighboring phases and very susceptible to small changes in the lattice and electronic structure. Yet direct access to the competing phases is still very difficult in FeTe and FeS because of the variation of the crystal quality across the families.

Here, we choose a slightly different approach and do not look directly at the electronic but rather at the lattice properties in FeS close to potential instabilities and use the Raman-active phonons as probes. We identify the A_{1g} and B_{1g} modes, a two-phonon scattering process, and a fourth mode from either defect-induced scattering or second-order scattering as well. These results are in good agreement with numerical calculations. Furthermore the temperature dependence of all phononic modes supports the results reported in Refs. [21,22], where emerging short range magnetic order at approximately 20 K was reported.

II. EXPERIMENT

Single crystals of FeS were synthesized as described elsewhere [23]. Before the experiment the samples were cleaved in air.

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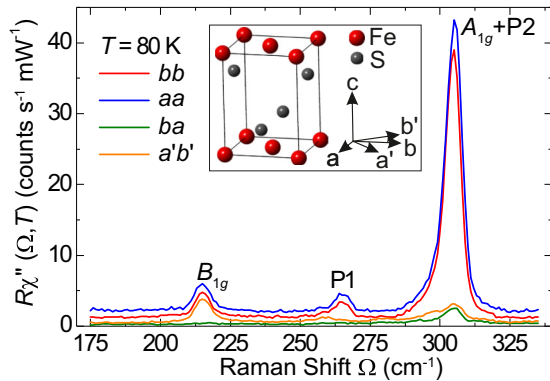


FIG. 1. Raman spectra of FeS at $T = 80$ K measured with light polarizations as indicated. The inset shows the crystal structure of FeS and the polarization directions with respect to the crystal orientation.

Calibrated customized Raman scattering equipment was used for the experiment. The samples were attached to the cold finger of a He-flow cryostat having a vacuum of approximately 5×10^{-5} Pa. For excitation we used a diode-pumped solid state laser emitting at 575 nm (Coherent GENESIS). Polarization and power of the incoming light were adjusted in a way that the light inside the sample had the proper polarization state and, respectively, a power of typically $P_a = 3$ mW independent of polarization. The samples were mounted as shown in the inset of Fig. 1. The crystallographic axes are a and b with $|a| = |b|$. The c axis is parallel to the optical axis. a' and b' are rotated by 45° with respect to a and b . The laser beam reached the sample at an angle of incidence of 66° and was focused to a spot of approximately $50 \mu\text{m}$ diameter. The plane of incidence is the bc plane. By choosing proper in-plane polarizations of the incident and scattered light the four symmetry channels A_{1g} , A_{2g} , B_{1g} , and B_{2g} of the D_{4h} space group can be accessed. Additionally, for the large angle of incidence, exciting photons being polarized along the b axis have a finite c -axis projection and the E_g symmetry can also be accessed. For the symmetry assignment we use the 2 Fe unit cell (crystallographic unit cell).

The observed phonon lines were analyzed quantitatively. Since the phonon lines are symmetric and $\Gamma_L(T) \ll \omega(T)$ the intrinsic line shape can be described by a Lorentz function with a central temperature dependent energy $\omega(T)$ and a width $\Gamma_L(T)$ (FWHM). The widths turn out to be comparable to the resolution σ of the spectrometer. Therefore, the Lorentzian needs to be convoluted with a Gaussian having width $\Gamma_G \equiv \sigma$.

III. THEORY

The electronic structure and the phonon dispersion were calculated using density functional theory (DFT) and density functional perturbation theory (DFPT), respectively, [24] within the QUANTUM ESPRESSO package [25]. The calculations were performed with the experimental unit cell parameters $a = 3.6735 \text{ \AA}$, $c = 5.0328 \text{ \AA}$, and $z = 0.2602$, where z is the height of the sulfur atoms above the Fe plane in units of the c axis [26]. We used the Vanderbilt ultrasoft pseudopotentials with the Becke-Lee-Yang-Parr (BLYP) exchange-correlation functional and s and p semicore states included in the valence for iron. The electron-wave-function and density energy cut-

offs were 70 Ry and 560 Ry, respectively, chosen to ensure stable convergence of the phonon modes. We used a Gaussian smearing of 0.01 Ry. The Brillouin zone was sampled with a $16 \times 16 \times 16$ Monkhorst-Pack k -space mesh. Our electronic structure and phonon calculations are in agreement with previously reported results [27,28].

The experimental positions of the S atoms entail a nonzero z component of the force of $6 \times 10^{-2} \text{ Ry}/a_B$ acting on them with a_B the Bohr radius. However, the relaxation of the z positions of the S atoms would result in a large discrepancy between the calculated and experimental energies of the optical branches [28], whereas the phonon frequencies calculated from experimental structure parameters are in good agreement with the experiment (see Table II). When using the measured lattice parameters, including atomic positions, some of the acoustic phonons are unstable and do not have a linear dispersion at small k . Upon relaxing the atomic positions the acoustic dispersion becomes linear and the energies at the zone boundary decrease slightly. The energies of the optical branches, on the other hand, increase by some 10%. Having all this in mind, we choose to use the experimental lattice parameters stated above. In this sense our calculations should be understood as a compromise.

The phonon dispersion and the density of states were calculated on a $6 \times 6 \times 6$ Monkhorst-Pack k -point mesh, and the dispersion is interpolated along the chosen line. The calculated phonon dispersions of the experimental and relaxed structures qualitatively coincide and display similar shapes and a gap. Discrepancies only appear in the absolute energies.

The selection rules for two-phonon processes were calculated using the modified group projector technique (MGPT) [29], which avoids summing over an infinite set of space group elements.

IV. RESULTS AND DISCUSSION

A. Polarization dependence

Raman spectra of FeS for four linear polarization configurations at a sample temperature of $T = 80$ K are shown in Fig. 1. Three peaks can be identified at 215, 265, and 305 cm^{-1} . The symmetric peak at 215 cm^{-1} shows up for aa , bb , and $a'b'$ polarizations, but vanishes for ba polarization. Hence the excitation obeys B_{1g} selection rules and can be identified as the out-of-phase vibration of iron atoms along the c axis. The strongest slightly asymmetric peak at 305 cm^{-1} obeys A_{1g} selection rules with contributions of order 5% in ba and $a'b'$ polarizations from either leakage or defect-induced scattering. An asymmetric Fano-type line shape can be acquired by coupling a phonon to an electronic continuum. However, as shown in Fig. 6 in the Appendixes, we find that the superposition of two symmetric, yet spectrally unresolved peaks gives a better agreement with the data than the description in terms of a Fano function. The stronger peak at 305 cm^{-1} has A_{1g} symmetry with some remaining leakage. We therefore identify this mode with the in-phase vibration of sulfur atoms along the c axis. The second peak, labeled P2, appears in spectra with parallel light polarizations and vanishes in ba , but has some contribution in $a'b'$ polarizations, suggesting mixed A_{1g} and B_{1g} symmetry. The third peak, labeled P1, is symmetric and appears only in

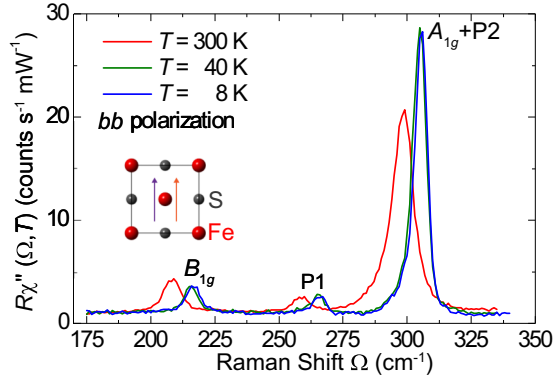


FIG. 2. Raman spectra of FeS in bb polarization projecting $A_{1g} + B_{1g} + E_g$ symmetries measured at temperatures given in the legend. The inset shows the light polarizations with respect to the crystal orientation.

spectra with parallel light polarizations and thus has pure A_{1g} symmetry.

B. Temperature dependence

For properly assigning all observed modes and for getting access to putative phase transitions we studied the temperature dependence. Figure 2 shows Raman spectra in bb polarization at 8, 40, and 300 K. The three peaks shift to higher energies upon cooling. The fourth peak P2 cannot be resolved in the raw data and can only be analyzed after a fitting procedure (see Appendix B). The peak energies $\omega(T)$ and the (intrinsic) linewidths $\Gamma_L(T)$ were determined as described at the end of Sec. II. All four modes show a monotonous increase in energy and decrease in linewidth upon cooling as shown in Fig. 3. Below 20 K the increase in the energies accelerates. We first address this overall behavior and disregard the anomaly around 50 K for the moment.

The shift and narrowing of all modes can be explained in terms of lattice contraction using a constant Grüneisen parameter γ and anharmonic decay into other phonon modes, respectively. The change in the (Lorentzian) linewidth $\Gamma_L(T)$ is given by [30]

$$\Gamma_L(T) = \Gamma_{L,0} \left(1 + \frac{2\lambda_{\text{ph-ph}}}{\exp\left(\frac{\hbar\omega_0}{2k_B T}\right) - 1} \right). \quad (1)$$

The zero temperature limits $\Gamma_{L,0}$ and ω_0 were obtained by extrapolating the respective experimental points of $\Gamma_L(T)$ and $\omega(T)$ in the range $20 \leq T \leq 50$ K to $T = 0$ (Fig. 3). With the phonon-phonon coupling $\lambda_{\text{ph-ph}}$ being the only free parameter the temperature dependence of $\Gamma_L(T)$ can be described as shown by red dashed lines in Fig. 3. The phonon energy $\omega(T)$ contains contributions from both the anharmonic decay and the lattice contraction, which depends essentially on the thermal occupation of the phonons, and can be written as [31]

$$\omega(T) = \omega_0 \left[1 - \gamma \frac{V(T) - V_0}{V_0} - \left(\frac{\Gamma_{L,0}}{\sqrt{2}\omega_0} \right)^2 \left(1 + \frac{4\lambda_{\text{ph-ph}}}{\exp\left(\frac{\hbar\omega_0}{2k_B T}\right) - 1} \right) \right]. \quad (2)$$

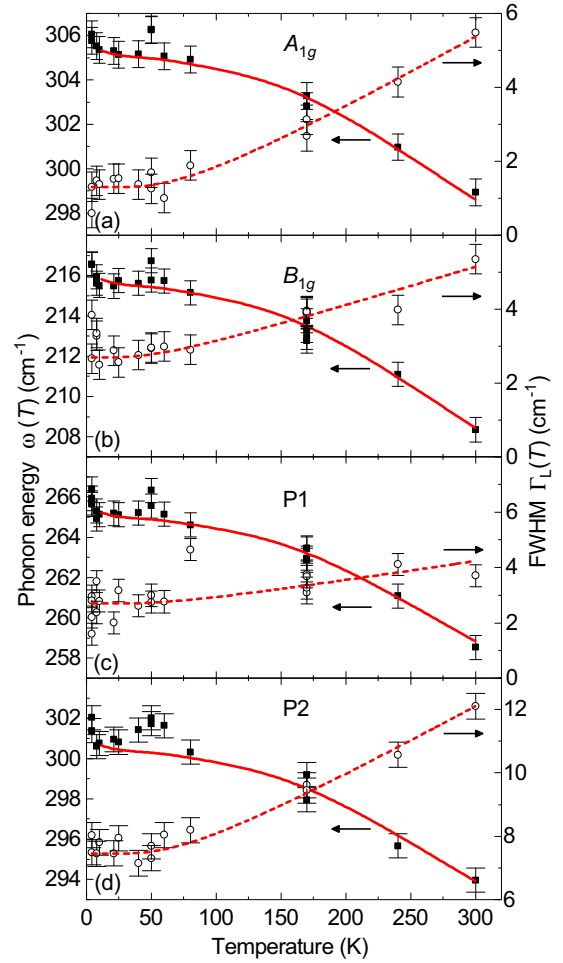


FIG. 3. Temperature dependence of energy and width of the four observed phonon modes in FeS. Black squares show the phonon energies ω ; open circles denote the phonon linewidths Γ_L . The red dashed and solid lines represent the temperature dependencies of the phonon linewidths and energies according to Eqs. (1) and (2), respectively. For better visualizing the low-temperature part, the data of this figure are plotted on a logarithmic temperature scale in Fig. 8 of Appendix D.

$V(T)$ and V_0 are the volumes of the unit cell at temperatures T and $T \rightarrow 0$, respectively. The numbers for the calculations are taken from Ref. [20]. The second term describes the effect of phonon damping on the line position in the harmonic approximation. Using $\lambda_{\text{ph-ph}}$ from Eq. (1), the Grüneisen parameter γ is the only free parameter and is assumed to be constant. The temperature dependencies $\omega(T)$ resulting from the fits are plotted in Fig. 3 as solid red lines. The numerical values for parameters γ and $\lambda_{\text{ph-ph}}$ obtained from the T -dependent energy and linewidth are compiled in Table I.

Below 20 K and around 50 K anomalies are found in the experimental data as follows:

(i) At 50 K the peak energies of all four modes deviate significantly from the otherwise smooth temperature dependence. The nearly discontinuous increase in energy could be reproduced for the A_{1g} phonon and peak P2 in multiple measurements. For the B_{1g} phonon and mode P1 the anomaly is not as clearly reproducible. The energy anomalies do not

TABLE I. Symmetry, Grüneisen constant γ , and phonon-phonon coupling parameter $\lambda_{\text{ph-ph}}$ of the four experimentally observed modes.

Mode	Symmetry	γ	$\lambda_{\text{ph-ph}}$
S	A_{1g}	2.2	1.68
Fe	B_{1g}	3.4	0.31
P1	A_{1g}	2.4	0.25
P2	$A_{1g} + B_{1g}$	2.2	0.31

have a correspondence in the linewidth. As there is neither an abrupt change in the lattice constants [20] nor any other known phase transition close to 50 K the origin of this anomaly remains unexplained although we consider it significant.

(ii) Upon cooling from 20 K to 4 K all four modes exhibit sudden, yet small, increases in energy. The changes in width are heterogeneous in that the A_{1g} mode narrows and the B_{1g} mode broadens. No clear tendencies can be derived for modes P1 and P2. Sudden changes in the temperature dependence typically indicate phase transitions. Yet, no phase transition has been identified so far. However, the anomaly at 20 K coincides with the emergence of short range magnetic order as inferred from two μSR studies [21,22]. Susceptibility measurements on a sample from the same batch were inconclusive. On the other hand, the XRD data show a small anomaly in the lattice parameters and the unit cell volume does not saturate at low temperature but rather decreases faster between 20 K and 10 K than above 20 K [20]. This volume contraction by and large reproduces the change in the phonon energies as can be seen by closely inspecting the low-temperature parts of Fig. 3 (see also Fig. 8). Hence the indications of short-range magnetism in FeS found by μSR have a correspondence in the temperature dependence of the volume and the phonon energies.

Clear phonon anomalies were observed at the onset of the spin density wave (SDW) phases in 122 systems [32–34] and of the more localized magnetic phase in FeTe [35], whereas continuous temperature dependence of the phonons was found in systems without long-range magnetism [36,37]. Upon entering the SDW state in the 122 systems the A_{1g} (As) mode softens abruptly and narrows by a factor of 3, whereas the B_{1g} (Fe) mode stays pinned and narrows only slightly [32]. The strong coupling of the As mode to magnetism was traced back to the interaction of the Fe magnetic moment with the Fe-As tetrahedra angle [38], which goes along with a change of the c -axis parameter. In Fe_{1+y}Te the roles of the B_{1g} and A_{1g} modes are interchanged [35,39,40]. In contrast, all four modes observed here in FeS harden below $T^* \approx 20$ K being indicative of a type of magnetic ordering apparently different from that in the other Fe-based systems.

Very recently, commensurate magnetic order with a wave vector of $\mathbf{q} = (0.25, 0.25, 0)$ was found in FeS below $T_N = 116$ K using neutron powder diffraction [41]. In the Raman spectra no anomalies can be seen around 120 K even if the range is studied with fine temperature increments of 10 K as shown in Appendix C. However, a small change in the temperature dependence of the c -axis parameter is observed around 100 K by XRD [20], which could be related to this type of magnetic order. Since the influence on the volume is small there is no detectable impact on the phonons.

TABLE II. Raman active phonon modes in t-FeS. Shown are the symmetries, the theoretical predictions for the experimental lattice parameters at $T = 0$, and the atoms involved in the respective vibrations. The experimental energies in the third column are extrapolations to $T = 0$ of the points measured between 20 K and 50 K.

Symmetry	Phonon energy (cm^{-1})		Atomic displacement
	Calculation	Experiment	
A_{1g}	316.1	305.3	S
B_{1g}	220.4	215.8	Fe
E_g	231.6		Fe, S
E_g	324.8		Fe, S

C. Analysis of the modes P1 and P2

Based on the energies, the selection rules, and the temperature dependence we first clarify the phononic nature of the two lines P1 and P2, which cannot as straightforwardly be identified as lattice vibrations as the in-phase sulfur and out-of-phase iron vibrations at 305.3 and 215.8 cm^{-1} . Second we derive their origin from the phonon density of states (PDOS) calculated for the zero-temperature limit.

All experimental energies for $T \rightarrow 0$ were derived from the points at low temperature as described in Sec. IV B (see also Fig. 3). The results for the modes at the Γ point are summarized in Table II and can be directly compared to the results of the calculations. The discrepancies between the experimental and theoretical energies for the Raman-active phonons are smaller than 4%. The price for this accuracy in the optical energies is an instability and possibly too high energies in the acoustical branches at small and, respectively, large momentum (see Sec. III).

The unidentified peaks P1 and P2 appear in the spectra measured with aa polarization, where none of the electric fields has a projection on the c axis. Thus they cannot have E_g symmetry obeying ca and cb selection rules. In addition, the observed energies would be relatively far off of the calculated energies (see Table II). Both peaks exhibit temperature dependencies similar to those of the two Raman-active phonons and the Grüneisen parameters are close to the typical value [42] of 2 and similar to those of the Raman-active phonons. The phonon-phonon coupling parameters $\lambda_{\text{ph-ph}}$ derived from the temperature dependence of the linewidths are close to 0.3 similar to that of the B_{1g} phonon. $\lambda_{\text{ph-ph}}$ of the A_{1g} phonon is roughly five times bigger for reasons we address later. Yet, because of the small prefactor $(\Gamma_{L,0}/\sqrt{2}\omega_0)^2 = O(10^{-3})$, the contribution of phonon-phonon coupling to the temperature dependence of $\omega(T)$ remains negligible in all cases and the phonon energies are essentially governed by the lattice contraction. These considerations demonstrate the phononic origin of the peaks P1 and P2.

In the second step we try to identify the phonon branches to which P1 and P2 can be related. To this end the full phonon dispersion and density of states (PDOS) were derived as described in Sec. III and are plotted in Fig. 4.

Independent of using the relaxed or experimental structure, P1 is located in the gap of the (theoretical) PDOS and cannot result from first order defect-induced Raman scattering. What

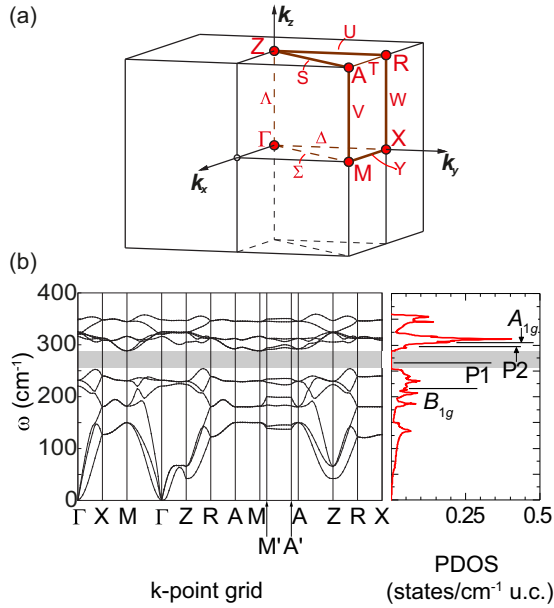


FIG. 4. Phonon dispersion of t-FeS. (a) Brillouin zone with high symmetry points and lines [43]. (b) Phonon dispersion along the directions as indicated and phonon density of states (PDOS). The gray-shaded area marks the gap in the phonon dispersion. The dispersion shown here is derived using experimental lattice parameters. For this reason some of the acoustic phonons are unstable and do not have a linear dispersion around the Γ point. Upon relaxing the structure the acoustic dispersion becomes linear at Γ , and the energies at the zone boundary decrease slightly. The energies of the optical branches, on the other hand, increase by some 10%. $M' = (0.4, 0.4, 0.0)$ and $A' = (0.4, 0.4, 0.5)$. The experimental energies of the four observed modes are shown as black lines.

alternatives exist for explaining P1? If we exclude exotic explanations such as a collective mode for the reasons given above the energy of $\omega_{P1} = 265 \text{ cm}^{-1}$ can only be obtained by the sum of two phonon modes having equal energy $\omega_{P1}/2$ and momenta \mathbf{k} and $-\mathbf{k}$ (for maintaining the $q \approx 0$ selection rule). As shown for various transition metal compounds including TiN, ZrN, or NbC second-order phonon Raman scattering can occur in the presence of defects [44]. Then first-order scattering being proportional to the PDOS (modulo energy and symmetry dependent weighting factors) is expected to be also substantial if not stronger. Although our crystals are slightly disordered there is no indication of substantial intensity at energies with high PDOS as can be seen by directly comparing Figs. 1 and 4(b). Alternatively, second-order scattering can originate in enhanced electron-phonon coupling [45]. In either case the energies of two phonons add up as they get excited in a single scattering process. Generally, no selection rules apply for second order Raman scattering and the resulting peak would appear in all symmetry channels [46]. Exceptions exist if the phonon wave vectors coincide with high-symmetry points or lines of the Brillouin zone.

From the phonon dispersion alone several phonon branches having \mathbf{k} and $-\mathbf{k}$ and energies in the range around $\omega_{P1}/2$ could add up to yield 265 cm^{-1} (see Fig. 4). However, as explained in Appendix F and shown in Table III for the space group P4/nmm

of t-FeS, the A_{1g} selection rules of P1 exclude all nonsymmetric combinations of branches (right column of Table III). On the other hand, all symmetric combinations include A_{1g} selection rules for the two-phonon peak (left column of Table III) and one has to look essentially for a high PDOS in the range $\omega_{P1}/2$. As shown in Fig. 4(b) the PDOS has a maximum in the right energy range. Since the maximum results from momenta away from the high-symmetry points or lines (see Fig. 4) which alone lead to pure A_{1g} symmetry one expects also intensity in B_{1g} and E_g symmetry as opposed to the experiment. For exclusive A_{1g} selection rules only seven possibilities exist. Since phase space arguments favor modes having a flat dispersion in extended regions of the Brillouin zone the Γ , M , and/or A points are unlikely to give rise to P1, and only the lines $S = A - Z$, $\Sigma = \Gamma - M$, and $V = A - M$ remain. The dispersion along the S or Σ branch contributes very little to the PDOS. On the high-symmetry line V a doubly degenerate branch would have a flat dispersion [see Fig. 4(b)] and contributes substantially to the PDOS but the energy of 150 cm^{-1} differs by 13% from the expected energy of 132.5 cm^{-1} . Instead of arguing about the accuracy of the theoretical phonon energies (see Sec. III) we looked at the dispersion close to but not strictly on V where the contribution to B_{1g} and E_g symmetries is expected to be still very small, e.g., along $M' - A'$ [Fig. 4(b)]. A detailed inspection shows that the maximum of the PDOS between 130 and 140 cm^{-1} comes from there. This explains both the selection rules and the energy of P1 to within a few percent.

Peak P2 cannot be explained in terms of one of the two E_g phonons either. As opposed to P1 it is not inside the gap of the PDOS and thus can originate from either first or second order scattering. If P2 originates in second order scattering in the same fashion as P1 there are five possibilities yielding $A_{1g} + B_{1g}$ but not E_g selection rules. As explained in the last paragraph only the branches $\Delta = \Gamma - X$ and $U = Z - R$ may contribute. For the low PDOS there we consider also first order defect-induced scattering for P2 to originate from. In fact, the PDOS possesses its strongest maximum 5 cm^{-1} below the (theoretical) A_{1g} phonon exactly where P2 is found. In spite of the very high PDOS here, the peak is weak explaining the negligible contributions from first order defect-induced scattering at lower energies. The high PDOS between 300 and 325 cm^{-1} may also be an alternative yet less likely explanation for the weak contributions in crossed polarizations in the energy range of the A_{1g} phonon (Fig. 1).

Finally, we wish to clarify whether the large phonon-phonon coupling $\lambda_{\text{ph-ph}}^{A_{1g}}$ found for the A_{1g} Raman-active mode (see Table I) is related to the appearance of P1. Due to the close proximity of the energies the A_{1g} mode apparently decays into states close to those adding up to yield P1. The decay is less restricted by symmetry leaving more options. For both processes the phonon-phonon coupling has to be substantial with the order of magnitude given by $\lambda_{\text{ph-ph}}^{A_{1g}} \approx 1.7$. Phonon-phonon coupling is present in any type of material because of the anharmonic potential. Defects enhance this effect [44]. Since FeS is a metal the phonon-phonon coupling goes at least partially through electronic states and may be indicative of enhanced electron-phonon coupling, $\lambda_{\text{el-ph}}$, as described, e.g., in Ref. [45]. The related contribution to $\lambda_{\text{ph-ph}}$ is then expected

TABLE III. Two-phonon processes in FeS. The symmetry group of the FeS system is the space group $P4/nmm$. For products of irreducible representations (IRs) in the left column Raman active modes (RM) in decomposition are given in the right one. Raman active modes of FeS are Γ_1^+ (A_{1g}), Γ_2^+ (B_{1g}), and two double degenerate Γ_5^+ (E_g). Γ_1^+ comes from vibrations of S atoms, Γ_2^+ from Fe ones, and both atom types contribute with one pair of Γ_5^+ modes. For complex representations ($V_{1,2,3,4}$ and all W) the double index indicates that the real representation is used, for example, $V_{13} = V_1 \oplus V_1^* = V_1 \oplus V_3$. Irreducible representations of the space group given in Ref. [53] are used.

Overtones		Combinations	
IR products (phonon states)	RM in decomposition	IR products (phonon states)	RM in decomposition
$[(\Gamma_i^\pm)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$\Gamma_1^h \otimes \Gamma_2^h, \Gamma_3^h \otimes \Gamma_4^h$ ($h = \pm$)	B_{1g}
$[(\Gamma_5^\pm)^2]$	A_{1g}, B_{1g}	$\Gamma_i^h \otimes \Gamma_5^h$ ($i = 1, 2, 3, 4, h = \pm$)	E_g
$[(X_i)^2]$ ($i = 1, 2$)	A_{1g}, B_{1g}, E_g	$X_1 \otimes X_2$	E_g
$[(M_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$M_1 \otimes M_2, M_3 \otimes M_4$	B_{1g}
$[(\Sigma_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$M_1 \otimes M_3, M_1 \otimes M_4, M_2 \otimes M_3, M_2 \otimes M_4$	E_g
$[(\Delta_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}, B_{1g}	$\Sigma_1 \otimes \Sigma_2, \Sigma_3 \otimes \Sigma_4$	B_{1g}
$[(V_{13})^2], [(V_{24})^2], [(V_5)^2]$	A_{1g}	$\Sigma_1 \otimes \Sigma_3, \Sigma_1 \otimes \Sigma_4, \Sigma_2 \otimes \Sigma_3, \Sigma_2 \otimes \Sigma_4$	E_g
$[(W_{13})^2], [(W_{24})^2]$	A_{1g}, B_{1g}, E_g	$\Delta_1 \otimes \Delta_2, \Delta_1 \otimes \Delta_3, \Delta_2 \otimes \Delta_4, \Delta_3 \otimes \Delta_4$	E_g
$[(Y_1)^2]$	A_{1g}, B_{1g}, E_g	$V_{13} \otimes V_{24}$	Γ_2^+
$[(Z_i^\pm)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$V_{13} \otimes V_5, V_{24} \otimes V_5$	Γ_5^+
$[(Z_5^\pm)^2]$	A_{1g}, B_{1g}	$W_{13} \otimes W_{24}$	Γ_5^+
$[(A_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$Z_1^h \otimes Z_2^h, Z_3^h \otimes Z_4^h$ ($h = \pm$)	B_{1g}
$[(R_i)^2]$ ($i = 1, 2$)	A_{1g}, B_{1g}, E_g	$Z_i^h \otimes Z_5^h$ ($i = 1, 2, 3, 4, h = \pm$)	E_g
$[(S_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$A_1 \otimes A_2, A_3 \otimes A_4$	B_{1g}
$[(U_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}, B_{1g}	$A_1 \otimes A_3, A_1 \otimes A_4, A_2 \otimes A_3, A_2 \otimes A_4$	E_g
$[(\Lambda_i)^2]$ ($i = 1, 2, 3, 4$)	A_{1g}	$R_1 \otimes R_2$	E_g
$[(\Lambda_5)^2]$	A_{1g}, B_{1g}	$S_1 \otimes S_2, S_3 \otimes S_4$	B_{1g}
$[(T_1)^2]$	A_{1g}, B_{1g}, E_g	$S_1 \otimes S_3, S_1 \otimes S_4, S_2 \otimes S_3, S_2 \otimes S_4$	E_g
		$U_1 \otimes U_2, U_1 \otimes U_3, U_2 \otimes U_4, U_3 \otimes U_4$	E_g
		$\Lambda_1 \otimes \Lambda_2, \Lambda_3 \otimes \Lambda_4$	B_{1g}
		$\Lambda_i \otimes \Lambda_5$ ($i = 1, 2, 3, 4$)	E_g

to be proportional to $\lambda_{\text{el-ph}}^2$. This conclusion is compatible with early results on the branch-dependent electron-phonon coupling in LaFeAsOF, where the strongest effects are reported for some Γ -point modes and the acoustic branches with intermediate to large momenta [47]. $\lambda_{\text{ph-ph}}^{A_{1g}} > 1$ and the two-phonon peak P1 indicate that the electron-phonon coupling is possibly larger than in the other Fe-based systems and reaches values up to unity. In BaFe₂As₂, as an example from the pnictide family, $\lambda_{\text{el-ph}}^2 \approx (1-4) \times 10^{-2} < \lambda_{\text{ph-ph}} \approx 0.1$ is reported [32,48,49]. On the other hand, one finds $\lambda_{\text{el-ph}}^2 \approx 0.4 < \lambda_{\text{ph-ph}} \approx 0.9$ for the E_g phonon in MgB₂, being generally believed to be a conventional superconductor [50,51]. Thus one may speculate whether $\lambda_{\text{el-ph}}$ might be even large enough in FeS to account for a T_c in the 5 K range.

V. CONCLUSION

We have studied and identified phonons in tetragonal FeS by Raman scattering. For the A_{1g} sulfur and B_{1g} iron mode the DFT and DFPT calculations agree to within a few percent with the experiment. A third observed peak within a gap in the theoretical phonon density of states can be identified as a second order scattering process involving two phonons. Both the selection rules, based on the modified group projector tech-

nique, and the energy are in agreement with the experiment. A fourth mode identified close to the A_{1g} sulfur phonon can be traced back to the biggest maximum of the PDOS and is most likely activated by a small amount of defects.

The temperature dependence of all four modes is governed by the contraction of the lattice, but shows anomalies at 50 K and below 20 K. The anomaly observed at 20 K has a correspondence in the thermal expansion [20] and μ SR experiments [21,22], which indicate short-range magnetic order. The long-range magnetic order observed recently by neutron diffraction experiments [41] below $T_N = 116$ K has no correspondence in the Raman spectra.

The appearance of two-phonon scattering indicates strong phonon-phonon scattering, which is likely to originate from an electron-phonon interaction being enhanced in comparison to other pnictides and chalcogenides. We argue that in FeS the T_c can in principle entirely result from electron-phonon interaction.

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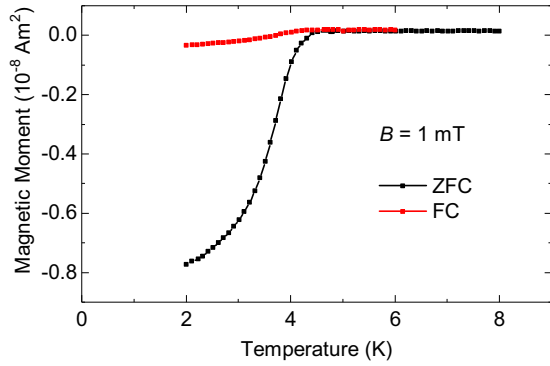


FIG. 5. Magnetization measurements of t-FeS at an applied field of $B = 1$ mT cooled to 2 K with (red curve) and without applied field (black curve).

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A.B. and A.M. have contributed equally to this work.

APPENDIX A: MAGNETIZATION MEASUREMENTS

Figure 5 shows magnetization measurements on a t-FeS sample from the batch studied in small applied fields. Measurements were done on a Quantum Design MPMS XL-7 SQUID magnetometer by cooling the sample to 2 K and sweeping

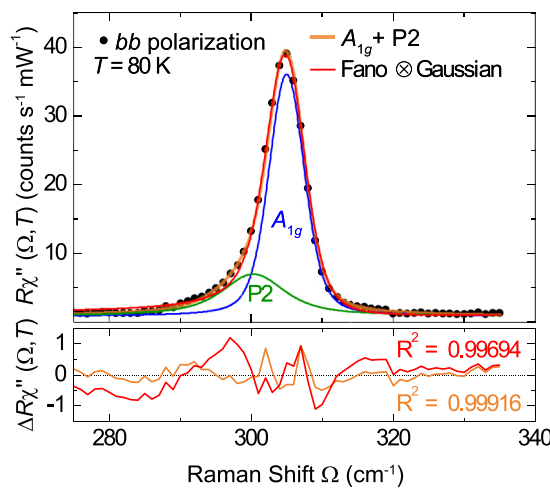


FIG. 6. Decomposition of the asymmetric phonon peak at 305 cm^{-1} . Measured data are shown as black dots. The orange line shows the sum of two Voigt profiles shown as blue and green lines, respectively. The convolution of Fano and Gaussian (red line) deviates in the peak flanks and the nearby continuum.

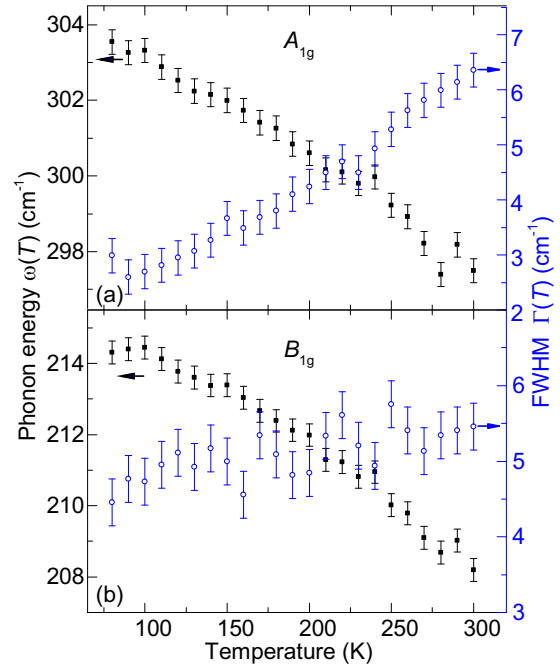


FIG. 7. Temperature dependence of A_{1g} and B_{1g} phonon modes in the temperature range between 80 K and 300 K. Black squares denote the phonon energies; open circles denote the phonon linewidths.

the temperature at 0.1 K/min. When cooled without applied field (ZFC, black curve) the sample shows a superconducting transition with onset at 4.5 K and a center of the transition at 3.6 K. When cooled in an applied field the magnetization decreases only weakly in the superconducting state indicating strong pinning.

APPENDIX B: DECOMPOSITION OF THE LINE AT 305 cm^{-1}

The peak at 305 cm^{-1} at low temperatures shows a significant asymmetry towards lower energies (see also Fig. 1). Coupling of the A_{1g} phonon mode to an electronic continuum by strong electron-phonon coupling would result in a line shape given by the convolution of a Fano function and a Gaussian, the latter representing the resolution of the spectrometer. We find, however, that this does not yield a satisfactory description of the measured line shape as can be seen from the red curve in Fig. 6, and thus conclude that the asymmetry of the peak stems from the overlap of two peaks which cannot be resolved separately. The corresponding line shape is the sum of two Lorentzians convoluted with a Gaussian which governs the resolution of the setup. Due to the distributivity of the convolution this is identical to the sum of two Voigt functions sharing the same width Γ_G of the Gaussian part. The overall spectral shape is shown in Fig. 6 as an orange line and agrees excellently with the data. The two contributing lines are shown in blue and green. From the selection rules (see Fig. 1) we identify the blue curve as the in-phase vibration of sulfur atoms in A_{1g} symmetry. The green line denotes a second mode P2, the origin of which is discussed in the main text.

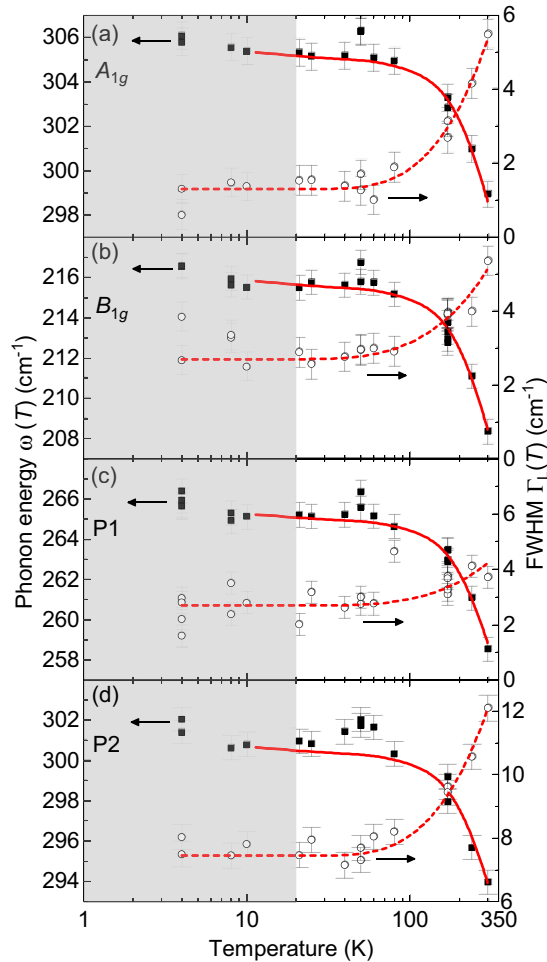


FIG. 8. Temperature dependence of energy and width of the four observed phonon modes in FeS on a logarithmic scale. The data is identical to Fig. 3 of the main text. Black squares show the phonon energies ω ; open circles denote the phonon linewidths Γ_L . The red dashed and full lines represent the temperature dependence of the phonon linewidths and energies according to Eqs. (1) and (2), respectively. The region below 20 K is shaded light gray. Since the data for the volume are limited to the range above 10 K the theoretical curves for the phonon energies (full red lines) end at 10 K.

APPENDIX C: DETAILED TEMPERATURE DEPENDENCE FOR $80 \leq T \leq 300$ K

Figure 7 shows the temperature dependence of the energies ω and linewidths $\Gamma(T)$ (FWHM) from 80 K to 300 K measured in temperature increments of 10 K. Raman scattering measurements were performed using a Jobin Yvon T64000 Raman system in micro-Raman configuration. A solid state laser with 532 nm line was used as an excitation source. Measurements were performed in high vacuum (10^{-6} mbar) using a KONTI CryoVac continuous helium flow cryostat with 0.5 mm thick window. Laser beam focusing was accomplished using a microscope objective with $\times 50$ magnification. The samples were cleaved right before being placed in the vacuum. As can be seen from Fig. 7, there is no deviation from the standard temperature behavior around 120 K.

APPENDIX D: TEMPERATURE DEPENDENCE ON A LOGARITHMIC SCALE

To better illustrate the behavior of the phonons at low temperatures Fig. 8 shows the experimental data and the theoretical curves from Fig. 3 of the main text on a logarithmic temperature scale. The region below 20 K is shaded light gray. As explained in Sec. IV B all four modes show an increase in energy below 20 K instead of the expected saturation, indicative of the putative onset of short range magnetic order. This effect manifests itself also in an incipient decrease of the unit cell volume [20] and is visible in the theoretical results for the phonon energies (full red lines). No clear tendency can be seen for the linewidths. The energy anomaly found around 50 K is discussed in the same section.

APPENDIX E: SECOND SAMPLE BATCH

Figure 9 shows Raman spectra on a t-FeS sample from a different batch (E256) taken at $T = 310$ K. The sample was oriented the same way as described in the main text. All three modes are visible for parallel light polarizations (*bb*), but vanish for crossed polarizations (*ba*), confirming the selection rules observed in the sample described in the main text. The inset shows magnetization measurements on a sample from batch E256 similar to the ones described in Appendix A. The superconducting transition sets in at 4.1 K.

APPENDIX F: SELECTION RULES FOR TWO-PHONON PROCESSES AND MGPT

In the multiphonon scattering process the system goes from an initial vibrational state (ground vibrational state) $|0,0,\dots\rangle$ to a final multiphonon state $|n_\mu, n_{\mu'}, \dots\rangle$, where n_μ is the number of phonons in the same state μ and μ stands for the entire set of quantum numbers (quasimomentum k , angular momentum quantum number m , etc.). For two-phonon processes the final vibrational state is the state with two phonons in the same quantum state (double-phonon or the first overtone state) or with two phonons in different states (combination state). The corresponding matrix element for

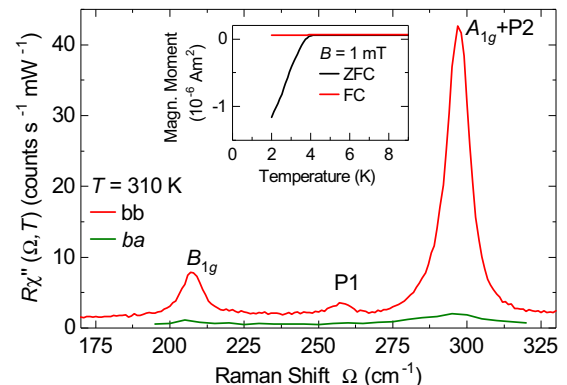


FIG. 9. Raman spectra of a t-FeS sample from a different batch taken at $T = 310$ K in polarizations as given in the legend. The inset shows magnetization measurements on a sample from this batch similar to Appendix A.

two-phonon Raman scattering is

$$\begin{aligned} &\langle 0, \dots, n_\mu, 0, \dots | \mathcal{R} | 0, 0, \dots \rangle, n_\mu = 2, \text{overtones,} \\ &\langle 0, \dots, n_\mu, 0, \dots, n_{\mu'}, \dots | \mathcal{R} | 0, 0, \dots \rangle, \\ &n_\mu = n_{\mu'} = 1, \text{combinations,} \end{aligned} \quad (\text{F1})$$

where \mathcal{R} is the Raman tensor. This matrix element should be a scalar or should transform as unit representation of the system space group \mathcal{S} . The standard approximation for the Raman tensor in infinite wavelength-light approximation for the non-resonant case is the polarizability tensor, which transforms as the (symmetrized) square of the vector representation, $D^{\mathcal{R}}(\mathcal{S})$. Decomposition of $D^{\mathcal{R}}(\mathcal{S})$ gives irreducible representations of the Raman active modes. The ground vibrational state transforms as unit representation, whereas the final two-phonon state transforms as symmetrized square, $[(D^\mu(\mathcal{S}))^2]$, of the corresponding irreducible representation $D^\mu(\mathcal{S})$ (overtones) or the direct product of two irreducible representations $D^\mu(\mathcal{S}) \otimes D^{\mu'}(\mathcal{S})$ (combinations). Symmetrization in the case of overtones comes from the bosonic nature of phonons. The matrix element [Eq. (F1)] transforms as reducible representation

$$\begin{aligned} &[(D^\mu(\mathcal{S}))^2] \otimes D^{\mathcal{R}}(\mathcal{S}), \text{ for overtones, or} \\ &D^\mu(\mathcal{S}) \otimes D^{\mu'}(\mathcal{S}) \otimes D^{\mathcal{R}}(\mathcal{S}), \text{ for combinations.} \end{aligned} \quad (\text{F2})$$

It is a scalar if the decomposition of the representations shown above contains the unit representation or, equivalently, if the intersection of decompositions of $[(D^\mu(\mathcal{S}))^2]$ or $D^\mu(\mathcal{S}) \otimes D^{\mu'}(\mathcal{S})$ and $D^{\mathcal{R}}(\mathcal{S})$ is a nonempty set. To obtain selection rules for two-phonon processes, following Birman's original method [52], it is enough to find the decomposition of $[(D^\mu(\mathcal{S}))^2]$ (for overtones) and $D^\mu(\mathcal{S}) \otimes D^{\mu'}(\mathcal{S})$ (for combinations) for all irreducible representations. If there is any representation

of the Raman active mode in those decompositions then that overtone or two-phonon combination is symmetrically allowed in the Raman scattering process. The decomposition of the (symmetrized) square of the vector representation is straightforward and is actually a finite dimensional point group problem. On the other hand, decomposition of $[(D^\mu(\mathcal{S}))^2]$ or $D^\mu(\mathcal{S}) \otimes D^{\mu'}(\mathcal{S})$ for any irreducible representation could be a difficult task because space groups are infinite. In the standard method based on character theory summation over all group elements is used and it is a problem in the infinite case. Therefore, it is necessary to apply a method which avoids summation over group elements. As is proven in Ref. [29] the modified group projector technique (MGPT) uses only group generators and finite dimensional matrices. Actually, the decomposition $D(\mathcal{S}) = \bigoplus_\mu f_D^\mu D^{(\mu)}(\mathcal{S})$ of the arbitrary reducible representation $D(\mathcal{S})$ into irreducible representations is effectively a determination of the frequency numbers f_D^μ . The MGPT expression for frequency numbers involves group generators s_i only:

$$f_D^\mu = \text{Tr} F \left(\prod_{i=1}^S F(D(s_i) \otimes D^{(\mu)*}(s_i)) \right). \quad (\text{F3})$$

Here S is the number of group generators, $F(X)$ is the projector on the subspace of the fixed points of the operator X , and Tr is the matrix trace (sum of the diagonal matrix elements). Consequently, the problem is reduced to calculation of the $S + 1$ projector to the fixed points. Technically, one looks for the eigenspaces for the eigenvalue 1 of each of the operators $D(s_i) \otimes D^{(\mu)*}(s_i)$, finding projectors on them, then multiplies the corresponding projectors, and repeats the procedure for the whole product from Eq. (F3). The trace of the final projector gives the corresponding frequency number.

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На основу члана 161 Закона о општем управном поступку и службене евиденције издаје се

УВЕРЕЊЕ

Милосављевић (Недељко) Ана, бр. индекса 2013/8010, рођена 30.06.1986. године, Смедерево, Смедерево-град, Република Србија, уписана школске 2017/2018. године, у статусу: финансирање из буџета; тип студија: докторске академске студије; студијски програм: Физика.

Према Статуту факултета студије трају (број година): три.
Рок за завршетак студија: у двоструком трајању студија.

Ово се уверење може употребити за регулисање војне обавезе, издавање визе, права на дечији додатак, породичне пензије, инвалидског додатка, добијања здравствене књижице, легитимације за повлашћену возњу и стипендије.

Овлашћено лице факултета



УНИВЕРЗИТЕТ У БЕОГРАДУ

ФИЗИЧКИ ФАКУЛТЕТ

Број 2642013

Београд, 30. 10. 2013. године

На основу члана 99. Закона о високом образовању (“Сл. гласник Републике Србије» број 76/05), и члана 9. и 184. Статута Физичког факултета (број 442/1 од 10.10.2006 и дате сагласности Универзитета у Београду број 02 612-1852 од 29.01.2007), у складу са Правилником о садржају и облику образаца јавних исправа које издају високошколске установе (“Сл. гласник Републике Србије» број 21/06, 66/06 и 8/07) издаје се следеће

У В Е Р Е Њ Е

Милосављевић (Недељко) Ана рођен-а 30. 06. 1986. године у Смедереву, Смедерево, Србија, уписан-а школске 2010/2011. године, завршио-ла је дипломске академске студије – студије другог степена (мастер) на студијском програму Физичког факултет Универзитета у Београду, смер: Теоријска и експериментална физика, дана 11. октобра 2013. године, са просечном оценом 9,33 (девет и 33/100) у току студија и постигнутим укупним бројем 60 ЕСПБ (шездесет ЕСП бодова) и тиме стекао-ла високо образовање и академски назив:

МАСТЕР ФИЗИЧАР - ТЕОРИЈСКА И ЕКСПЕРИМЕНТАЛНА ФИЗИКА- master

Уверење се издаје на лични захтев, а служи као доказ о завршеној стручној спреми до издавања дипломе.



Д Е К А Н

Проф. др Јаблан Дојчиловић

C 403861

ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

ТЕОРИЈСКА МЕХАНИКА

(назив предмета)
 риференцијелни испити

(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Испит	9	22. VII 2011	P. Zvezda-Hard
Закључна оцена	9	(1 револ)	(P. Zvezda-Hard)
Наставник	РАМАНОВА ДНЕКТРОСКОПИЈА		
(назив предмета) 15			
(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)			
Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Испит		28. 12. 2015.	
Закључна оцена	10	(1 револ)	(P. Zvezda-Hard)
Наставник			

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ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

Физику РАСТА КРИСТАЛУ

(назив предмета)
15

(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Испит	100	22. 01. 2016	(P. Zvezda-Hard)
Закључна оцена	10	(1 револ)	(P. Zvezda-Hard)
Наставник	НУКЛЕАРНА ФИЗИКА		
(назив предмета)			
(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)			
Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Испит	88	9. 6. 16	
Закључна оцена	9	(1 ДЕВЕТ)	(P. Zvezda-Hard)
Наставник			

ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

ЕНЕРГЕТИКА I

Идејна пројекат испис

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника	(шифра предмета) (ЕСПБ бодова)		(обавезни, изборни)	
Испит		30.09.2016		7			
Закључна оцена							
Наставник							
РАД НА ДОКТОРАТУ (I део) (назив предмета)							
Назив предиспитне обавезе							
Број поена							
Датум овере/полагања							
Потпис наставника							
Испит							
Закључна оцена (УРАДИМ)							
Наставник (Kovacevic)							

ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

ИЗЈАВЉЕ ЕКСПЕРИМЕНТАЛНЕ МЕТОДЕ

15

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника	(шифра предмета) (ЕСПБ бодова)		(обавезни, изборни)	
Испит		31.01.2017		10			
Закључна оцена							
Наставник							
ЕКСПЕРИМЕНТАЛНЕ МЕТОДЕ (назив предмета)							
Назив предиспитне обавезе							
Број поена							
Датум овере/полагања							
Потпис наставника							
Испит							
Закључна оцена (10)							
Наставник (Kovacevic)							

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ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

ФИЗИКА ЧВРАТОТ СТАЊК
(назив предмета)

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
		12.06.2017	
Испит			
Закључна оцена	10	(055BT)	
Наставник	Miki M. Mikić		
(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)			
Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Семини	10/10		
индоквизи	17/20		
испити	18/30		
честени	32/40		
Испит	74/100	12.9.2014.	
Закључна оцена	9	(055BT)	
Наставник	Miki M. Mikić		

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ПОДАЦИ О ПРЕДИСПИТНИМ ОБАВЕЗАМА И ПОЛАГАЊУ ИСПИТА

РАД НА ДОКТОРАТУ (II ДЕО)
(назив предмета)

Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Испит	85/100	5.2.2018.	
Закључна оцена	9	(155BT)	
Наставник	Miki M. Mikić		
(шифра предмета) (ЕСПБ бодова) (обавезни, изборни)			
Назив предиспитне обавезе	Број поена	Датум овере/полагања	Потпис наставника
Методи	42/50		
Испит	85/100	5.2.2018.	
Закључна оцена	9	(155BT)	
Наставник	Miki M. Mikić		



ДОКТОРСКЕ СТУДИЈЕ

ПРЕДЛОГ ТЕМЕ ДОКТОРСКЕ ДИСЕРТАЦИЈЕ
КОЛЕГИЈУМУ ДОКТОРСКИХ СТУДИЈА

Школска година
2017/2018

Подаци о студенту

Име

Ана

Презиме

Милосављевић

Број индекса

8010/2013

Научна област дисертације

Физика кондензоване материје и статистичка
физика

Подаци о ментору докторске дисертације

Име

Ненад

Презиме

Лазаревић

Научна област

Физика кондензоване материје и
статистичка физика

Звање

Виши научни сарадник

Институција

Институт за Физику

Предлог теме докторске дисертације

Наслов

Раманова спектроскопија суперпроводника на бази гвожђа

Уз пријаву теме докторске дисертације Колегијуму докторских студија, потребно је приложити следећа документа:

1. Семинарски рад (дужине до 10 страница)
2. Кратку стручну биографију писану у трећем лицу јединине
3. Фотокопију индекса са докторских студија

Датум	<input type="text" value="12.01.2018."/>	Потпис ментора	<input type="text"/>
		Потпис студента	<input type="text"/>

Мишљење Колегијума докторских студија	
Након образложења теме докторске дисертације Колегијум докторских студија је тему	
прихватио <input checked="" type="checkbox"/>	није прихватио <input type="checkbox"/>
Датум	Продекан за науку Физичког факултета
<input type="text" value="24.01.2018"/>	<input type="text" value=""/>